



Ultrasound assisted synthesis of polythiophene/SnO₂ hybrid nanolatex particles for LPG sensing



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ABSTRACT

Polythiophene (PTP) coated SnO₂ nano-hybrid particles have been synthesized using an ultrasound assisted in situ oxidative polymerization of thiophene monomers. Reference experiments have also been performed in the absence of ultrasound to clearly illustrate the effect of ultrasonic irradiations. FTIR results show broadening and shifting of peaks toward lower wave numbers, suggesting better conjugation and chemical interactions between PTP and SnO₂ particles. Due to strong synergetic interaction between the SnO₂ nanoparticles and polythiophene, this hybrid nanocomposite has the potential application as chemical sensors. It has been observed that PTP/SnO₂ hybrid sensors could detect liquefied petroleum gas (LPG) with high sensitivity at room temperature. PTP/SnO₂ hybrid composite containing 20 wt% SnO₂ showed the maximum sensitivity at room temperature. The sensing mechanism of PTP/SnO₂ hybrid nanocomposites to LPG was mainly attributed to the effects of p–n heterojunction between PTP and SnO₂.

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1. Introduction

Organic–inorganic hybrid core–shell nanoparticles are being used in number of engineering applications due to versatile functional properties of the organic and inorganic constituents. These core–shell particles often exhibit significant improvement in magnetic, optical and electrical properties, which make them attractive for both scientific and technological applications in the different areas such as catalysis, coatings, drug delivery, optics, electronics, biology etc. [1–3]. Encapsulation of the inorganic materials by the various polymer matrices, or the immobilization of the inorganic phase on the surface of the polymer particles can be used to prepare core–shell morphology of the nanocontainers. Immobilization can be physical or chemical in nature leading to varying extents of bonding strength between the two different phases. To achieve the compatibility of phases, inorganic particles are pre-dispersed in the continuous phase. The ultrasound assisted miniemulsion chemical oxidative polymerization in the presence of inorganic particles

leads to the formation of hybrid particles if the interaction between the two phases is controlled. The conditions of ultrasound assisted polymerization process can be optimized for achieving a nanoscale dispersion containing inorganic particles and these hybrid functional nanomaterials could be utilized for specific applications [4].

Ultrasonically induced cavitation can generate local temperatures as high as 5000 K and local pressures as high as 1000 atm, with local heating and cooling rates greater than 10⁹ K/s [5]. In this environment, water molecule dissociates into primary hydrogen radicals (H[•]) and hydroxyl radicals (OH[•]) in a collapsing cavitation bubble. Subsequently, these active species take part in driving the polymerization reactions. Also, ultrasonic irradiations help in creating the hybrid nanoparticles with their synergetic or complementary behaviors that are not observed in their single counterparts. The benefits of the ultrasound include narrow size distribution, high purity, unique reaction effects, higher yields, and the ability to form nanoparticles with uniform shapes [6–8]. Because of its ability to produce the hybrids with significantly beneficial properties, the use of ultrasound has shown a rapid growth in material science. These novel surface properties of the hybrid can play a crucial role in gas sensing applications and hence the present work has concentrated on synthesis of a hybrid material with possible potential for gas sensing application.

There have been some earlier studies reporting the use of hybrid materials for gas sensor applications. The

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drawbacks of conventional synthesis methods like higher reaction temperature, lower kinetic reaction rates, higher amount of initiator, low yields and selectivity can be overcome by the use of ultrasound due to the enhanced physical and chemical effects of acoustic cavitation in the heterogeneous systems [9–14]. Ram et al. [15] prepared polyhexythiophene/SnO₂ and poly(ethylenedioxythiophene)/SnO₂ composite thin films, which showed an excellent sensitivity to NO₂ at concentration of ppb levels. It was reported that the hybrid materials exhibit much higher sensitivity and better selectivity as compared to the pure inorganic and organic materials. Dhawale et al. [16] found that the polyaniline/TiO₂ sensor presented maximum response of 63% upon exposure to 0.1 vol% liquefied petroleum gas (LPG) in air at room temperature. Nicho et al. [17] developed a polyaniline composite sensor useful for detection of low concentrations of NH₃ gas. Hosono et al. [18] prepared intercalated polyaniline/MoO₃ hybrids, which showed good sensing characteristics for the VOCs. Many authors have investigated gas sensing properties of conducting polymer-SnO₂ hybrid composites for different gases like acetone, ethanol, methanol, NH₃, NO_x and hydrogen sulfide. However, there is not a single report about application of thiophene-SnO₂ hybrid composite for LPG sensing at room temperature. LPG is highly flammable and even at low level concentration (ppm), it becomes a serious threat to human lives. Due to this and significant usage in cooking/automobile application, it is imperative that fast and selective detection of LPG is necessary to avoid any leakage of gas, which can help in preventing the occurrence of accidental explosions.

In the present work, core shell hybrid materials have been prepared with SnO₂ nanoparticles as the core and polythiophene (PTP) as the shell using an ultrasound assisted in situ chemical oxidative polymerization. The cavitating conditions like local intense micromixing, high temperature/pressure pulse can be very helpful in getting the desired properties for the synthesized nanoparticles [19–25]. Initially SnO₂ nanoparticles have been synthesized in the presence of ultrasonic irradiations and later these have been used for preparation of hybrid material again in the presence of ultrasound. XRD and FTIR analysis have been performed to demonstrate possible synergetic interaction between PTP and SnO₂ nanoparticles. Gas sensing properties of the synthesized hybrid material to LPG have been systematically studied. The possible sensing mechanism of PTP/SnO₂ hybrid has also been discussed.

2. Materials and methods

2.1. Materials

Tin tetrachloride (SnCl₄·5H₂O) and urea ((NH₂)₂CO) of analytical grade and reagent grade ferric chloride were procured from Central Drug House, India. Thiophene was procured from Merck whereas chloroform (CHCl₃) and methanol (CH₃OH) were obtained from Sigma Aldrich. All the chemicals were used as received from the supplier.

2.2. Synthesis of SnO₂ nanoparticles in the presence of ultrasound

In the preparation of SnO₂ nanoparticles, 0.1 M aqueous solution of tin tetrachloride and urea has been utilized as starting materials. Synthesis was carried out by mixing tin tetrachloride and urea solutions in a ratio of 1:2 using ultrasonic horn with an operating frequency of 22 kHz and rated output power of 750 W. The mixture was irradiated for 1 h on 5 s pulse mode (5 s on and 5 s off mode). The standard depth of immersion of the horn in the solution was 1 cm and it was located at the center of the glass reactor to achieve the maximum cavitation [26]. The system was operated with 50% amplitude, giving theoretical energy dissipation rate as

375 W (0.50 × 750 = 375 W). The calorimetric efficiency of the ultrasonic horn, as measured in the current work, was about 12.8% giving actual energy dissipation in the liquid as 38.4 W. The white precipitate was separated by centrifugation at 6000 rpm for 30 min. Further, the final product was washed with distilled water several times and dried in an oven at 90 °C. The obtained SnO₂ nanoparticles were ground using mortar and pestle and finally calcined at 600 °C for 2 h.

2.3. Synthesis of SnO₂ encapsulated polythiophene hybrid via ultrasound assisted in situ chemical oxidative polymerization

The PTP/SnO₂ hybrids were prepared via ultrasound assisted in situ chemical oxidative polymerization using anhydrous FeCl₃ as the oxidant in batch mode (Fig. 1). The experimental setup consists of a reactor which is mainly a jacketed glass vessel of 400 mL volume capacity equipped with a 13 mm stainless steel probe connected to an ultrasonic generator.

In a typical synthesis, 0.052 g of SnO₂ was uniformly dispersed in 100 mL of chloroform in the reactor equipped with ultrasonic horn [27,28]. The irradiation time was 20 min in order to keep them in dispersed form in the solution. Then, 1 mL of thiophene monomer was added drop wise in the reactor and the mixture was sonicated for 20 min to achieve adsorption of the thiophene monomers on the surface of SnO₂. Further, oxidant FeCl₃ solution with a mole ratio of FeCl₃/Thiophene (2:1) was added, followed by sonication for 40 min. Reaction mixture was heated after addition of FeCl₃, and the temperature was maintained at 40 °C (±1 °C) throughout the duration of the experiment. The complete reaction was carried over a period of 60 min in the presence of ultrasonic irradiation to ensure complete encapsulation of the SnO₂ in the polymer latex. The color of the mixture changes from gray to black. The product was filtered and washed with methanol several times. During washing, the color of the product changed from black to red [29]. The final product was dried at 60 °C under vacuum for 12 h. By the same method, a series of PTP/SnO₂ hybrids varying in loading of SnO₂ were prepared and designated as PTP/(5%)SnO₂, PTP/(10%)SnO₂, PTP/(15%)SnO₂ and PTP/(20%)SnO₂. Also pure PTP was synthesized in the absence of SnO₂ by a similar procedure. Synthesis of hybrid PTP/SnO₂ nanocomposite with different loading of SnO₂ was also prepared via the conventional approach (in the absence of ultrasound) of chemical oxidative polymerization with an objective of establishing the role of ultrasound.

2.4. Characterization

X ray diffraction (XRD) pattern of pure SnO₂ and PTP/SnO₂ was recorded using powder X-ray diffractometer (XRD, Philips Xpert Pro PW-3040). The size and morphology of the nanoparticles of SnO₂ embedded into the polythiophene matrix was determined using a transmission electron microscope (TEM, JEOL JSM2000-EX II). An infrared (IR) spectrum of samples was also obtained using a SHIMADZU 8400S FTIR spectrometer in KBr medium at room temperature in the region of 4000–500 cm⁻¹. Particle size distribution analysis of SnO₂ encapsulated polythiophene was carried out by Photon Correlation Spectroscopy (PCS, 3000 HAS, analyzer Malvern). The weight loss of pure SnO₂ and hybrid nanocomposite was determined in the range of 30 to 600 °C using thermogravimetric analysis (PerkinElmer TGA system, USA), at a heating rate of 10 °C/min.

2.5. Gas sensor fabrication using PTP/SnO₂ hybrid

The PTP/SnO₂ hybrid was pressed into pellet form of diameter ~1.4 cm, thickness ~0.1 cm and the electrical contact leads were fixed at a distance of 1 cm with the help of silver paste on the

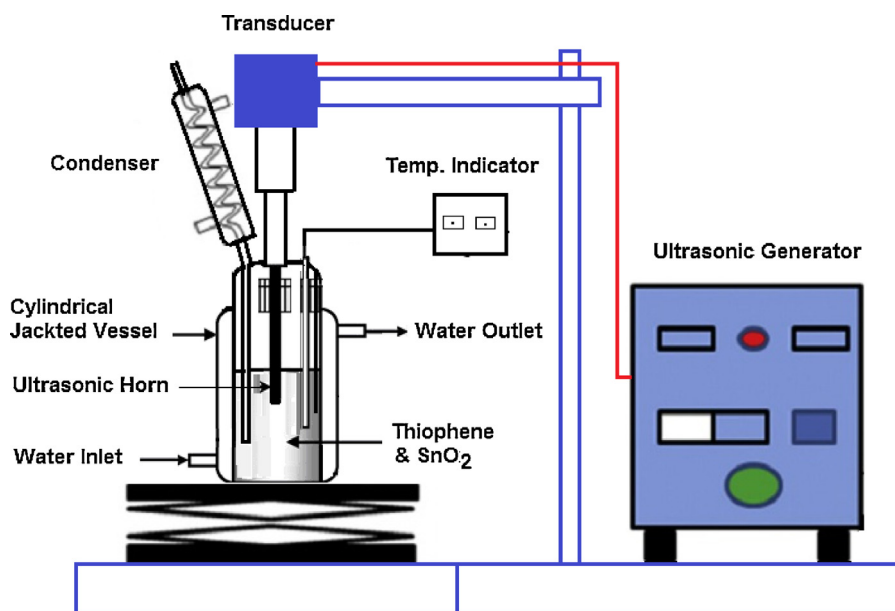


Fig. 1. Experimental setup for synthesizing polythiophene coated SnO₂ nano hybrid via ultrasound assisted chemical oxidative polymerisation.

surface of the pellet to form sensing element. LPG sensing experiments were carried out in a static gas chamber under ambient conditions. The sensing element was kept on a glass base which holds the sensing element tightly. The sensing measurements were performed by exposing the sensor (as pellets) to LPG in a closed glass dome (25 L) using a two probe method. The known volume of the LPG was introduced into the gas chamber filled with air and it was maintained at atmospheric pressure. Gas sensing setup comprises a temperature controller, K type thermocouple, an electrically heated plate, gas chamber and volume measurement unit [30]. The performance of the sensing element is presented in terms of the gas response (S), which is defined by the relation:

$$S = \frac{(R_g - R_a)}{R_a} \times 100$$

where R_a and R_g are values of the sensor resistance in air and in the presence of LPG, respectively.

3. Results and discussion

3.1. Mechanism of formation of PTP/SnO₂ hybrid using ultrasound assisted in situ chemical oxidative polymerization

PTP coated SnO₂ nanoparticles were successfully synthesized using ultrasound assisted in situ polymerization of thiophene monomers in chloroform using anhydrous FeCl₃ (oxidant and dopant) [31,32,24]. The process of formation of hybrid material is illustrated in Fig. 2. Initially, the reaction media containing SnO₂ nanoparticles, chloroform and thiophene is sonicated in order to adsorb thiophene monomer molecules on the surface of SnO₂ nano particles. After introducing anhydrous FeCl₃ into the reactor in the presence of ultrasound, polymerization of thiophene is initiated. In the initiation stage, radicals are generated through deprotonation of cation which is formed by the complexation between the thiophene sulfur, FeCl₃ and water molecules due to the cavitation effects generated by acoustic cavitation [33]. These free radicals enter the thiophene monomer droplets resulting into polymerization process. The radicals also accelerate the rate of thiophene polymerization and conversion. Due to ultrasound cavitation fast dissociation of the oxidant molecule takes place. Uniform size monomer droplets are formed due to violent shearing action

produced by acoustic cavitation. The physical effects of intense turbulence and liquid circulation currents due to cavitation results into a rapid dispersion of nano-sized SnO₂ into the monomer droplets [6].

During polymerization, SnO₂ nano particles are surrounded by oxidized polythiophene [34]. The polythiophene shell becomes thin due to the extraction of the iron chloride. During this process, the color changes from gray (oxidized state) to red (reduced state). Complete removal of gray color was achieved by the filtration of product and washing with methanol several times. During washing, the color of the product changed from black to red.

3.2. FTIR analysis of SnO₂ and PTP/SnO₂ hybrid nanocomposites

Fig. 3 shows the FTIR spectra of SnO₂ nanoparticles, pure PTP and PTP/SnO₂ hybrid material. For pure PTP, there are various weak peaks in the range of 2800–3100 and 1690 cm⁻¹, which can be attributed to the C–H stretching vibrations and C=C characteristic peak, respectively. In the finger print region of PTP, the IR absorption peak at 787 cm⁻¹ was recognized due to the C–H out of plane stretching vibration of the 2,5-substituted thiophene ring, which confirmed the polymerization of thiophene monomer. The absorption peak at 698 cm⁻¹ is assigned to the C–S bending mode, which indicates the presence of thiophene monomer [35,36]. The infrared spectra of PTP/SnO₂ hybrid show almost identical wave numbers and positions in the range of 500–4000 cm⁻¹ compared to that of the pure PTP, but in Fig. 3(b), the peak intensity reduced and a new peak appeared at around 700 cm⁻¹ demonstrating the presence of SnO₂ [37]. It is clear that, as compared to the peak at 685 cm⁻¹ reported in Fig. 3(a) of SnO₂, the peak of the hybrid is shifted to higher wavenumber (700 cm⁻¹). It clearly indicates that strong synergic interaction exists between SnO₂ and sulfur atom of the thiophene linkage [38,39]. The broadening and changes in peak positions suggest that a strong chemical interaction between PTP and SnO₂ is responsible to change the polymer conformation. It is expected that the presence of ultrasound may play a vital role in better dispersion of nano SnO₂ particles into polythiophene, creating a strong adhesion [40–42].

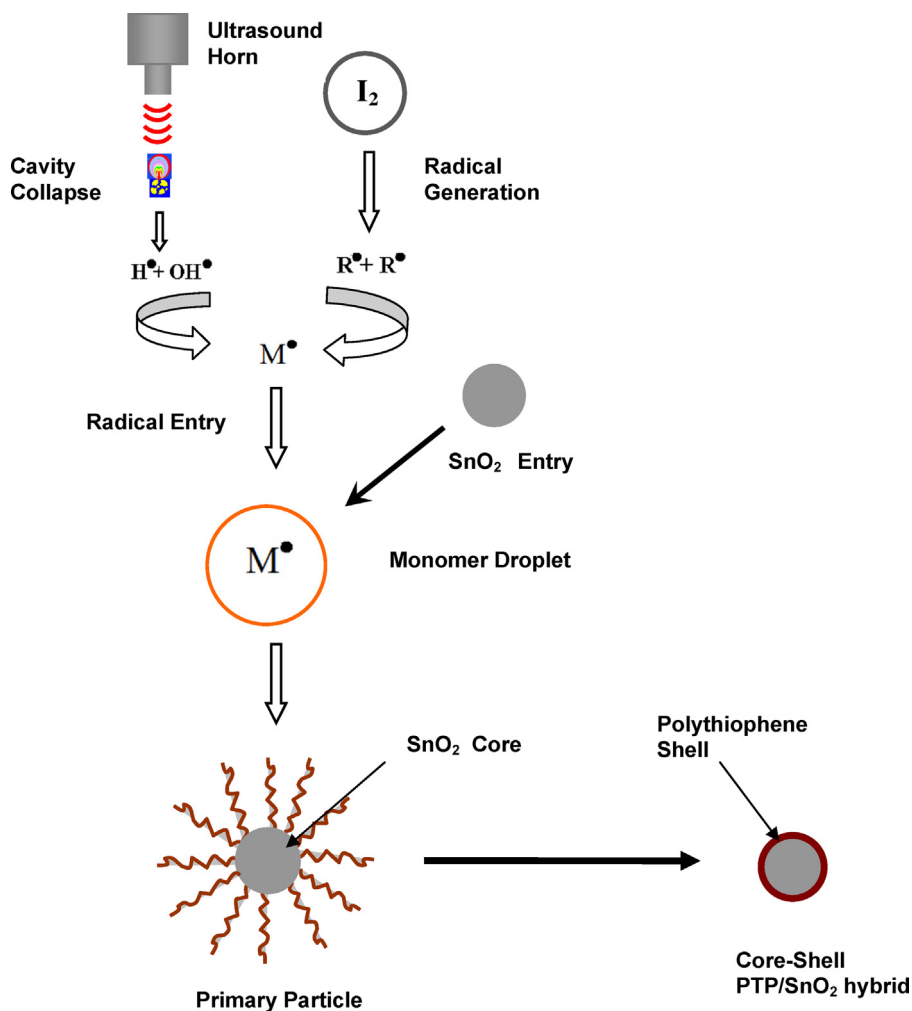


Fig. 2. Schematic illustration of the formation of polythiophene coated SnO₂ nano hybrid via ultrasound assisted chemical oxidative polymerisation.

3.3. X-ray diffraction analysis and TEM analysis of SnO₂ and PTP/SnO₂ hybrids

Fig. 4 shows the XRD patterns of the SnO₂, pure PTP and PTP/(10%) SnO₂ hybrid materials. All the peaks appeared in SnO₂ XRD diffractogram match well with the standard SnO₂ tetragonal rutile structure (JCPDS:41-1445). For PTP, the XRD pattern showed a broad, amorphous diffraction peak at approximately $2\theta = 20^{\circ}$ – 24° as depicted in Fig. 4(a). It indicates the existence of polythiophene, which is in accordance with the previous report [43]. The peak centered around 22.8° , corresponds to the intermolecular π – π stacking for the pure thiophene [44]. For PTP/(10%) SnO₂ hybrids, all the major peaks existing in the pure SnO₂ were also observed, but the peaks were weaker than those of the SnO₂, which may have resulted from the possible interaction between PTP and SnO₂ [45]. After addition of SnO₂ in the nanocomposite, the XRD peak gets shifted from 24° to 30° . It is attributed to induced strain in the polymer nanocomposite due to incorporation of SnO₂ in polythiophene by ultrasound assisted in situ chemical oxidative polymerization method.

The morphology of the SnO₂ and PTP/SnO₂ hybrid prepared by ultrasound assisted chemical oxidative polymerization is reported in Fig. 5(a) and (b) respectively. The diameter of the SnO₂ nanoparticles is about 15–20 nm. The TEM image of the PTP/SnO₂ hybrid shows that all SnO₂ nanoparticles are encapsulated by PTP, and no uncovered SnO₂ nanoparticles are observed. The SnO₂ nanoparticles are dispersed uniformly in PTP, which can potentially be

attributed to the ultrasonic irradiation used during polymerization, which may break the aggregation of SnO₂ nanoparticles. The problem of effective dispersion of SnO₂ nanoparticles into the polythiophene arising from their very high surface energy is expected

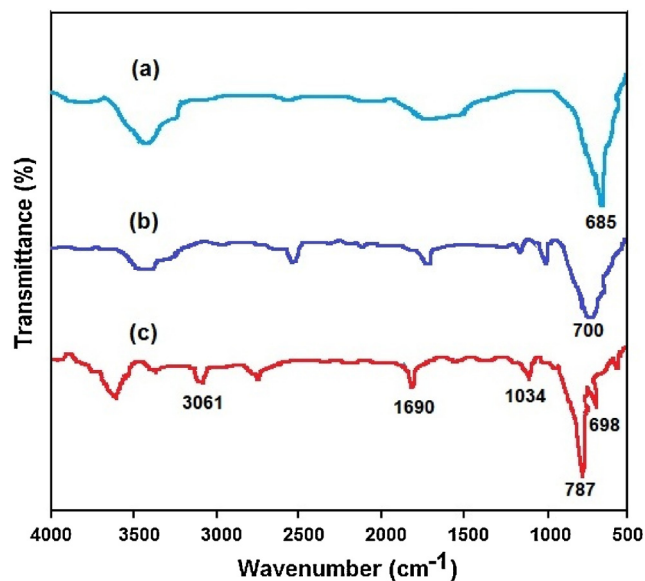


Fig. 3. FTIR spectra of nano size (a) SnO₂, (b) PTP/SnO₂ hybrid and (c) pure PTP.

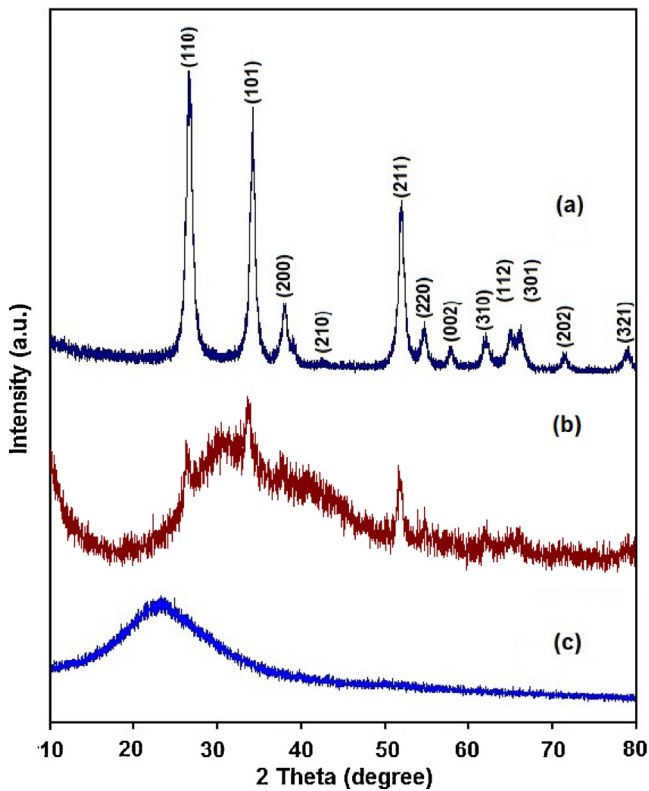


Fig. 4. X-ray diffraction of nano size (a) SnO_2 , (b) PTP/ SnO_2 hybrid and (c) pure PTP.

to be resolved by ultrasound as has been demonstrated for similar systems [46,47,24,48,49]. The formation of PTP encapsulated SnO_2 is due to the strong interaction between PTP and SnO_2 . It is expected that due to ultrasound, uniform and homogeneous distribution of SnO_2 can be obtained and hence the ultrasonically synthesized hybrid composite is expected to give an excellent response [50,51].

The particle size distribution of PTP/ SnO_2 hybrid prepared by ultrasound assisted and conventional chemical oxidative polymerization technique (in the absence of ultrasound) has been depicted in Fig. 6(a) and (b). The particle size of PTP/ SnO_2 hybrid nanoparticles prepared using ultrasound assisted approach was found to be in the range 60–80 nm which is about 10 times lower as compared to that observed (over the range 659–1177 nm) in the case of conventional chemical oxidative polymerization method (without use of ultrasound). The average particle size of PTP/ SnO_2 hybrid

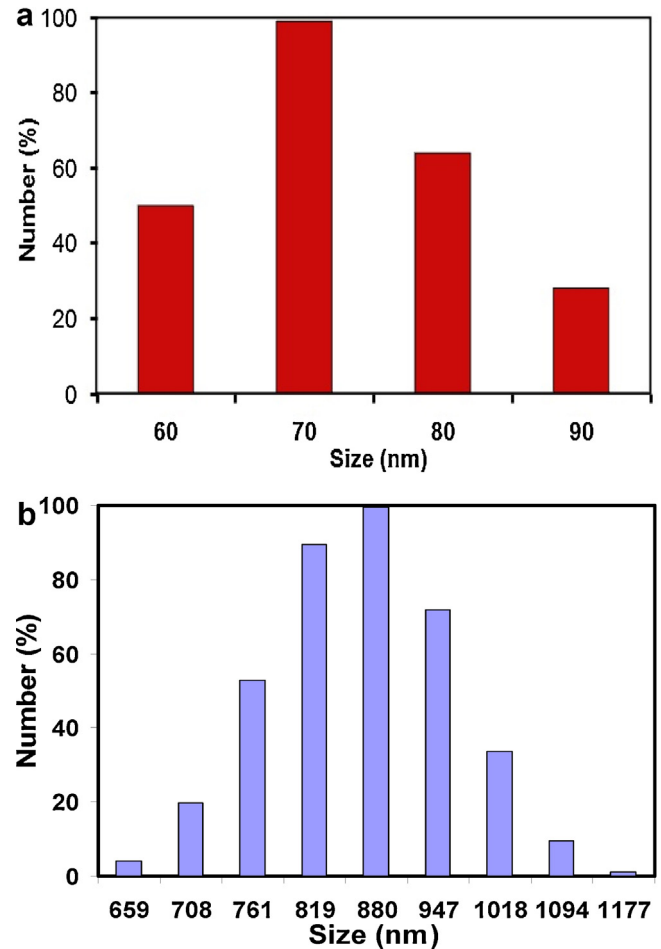


Fig. 6. (a) Particle size analysis of PTP/ SnO_2 nano hybrid prepared by ultrasound assisted chemical oxidative polymerization. Particle size analysis of PTP/ SnO_2 hybrid prepared via conventional chemical oxidative polymerization.

nanoparticles was 70 nm in the case of ultrasound assisted method whereas it was 834 nm for conventional chemical oxidative polymerization. The significant decrease in the particle size of PTP/ SnO_2 hybrid nanoparticles is attributed to the enhanced micromixing, rapid nucleation, improved solute transfer rate and formation of large number of nuclei due to physical effects of the ultrasonic irradiation [12]. Due to the fast kinetics of ultrasound assisted

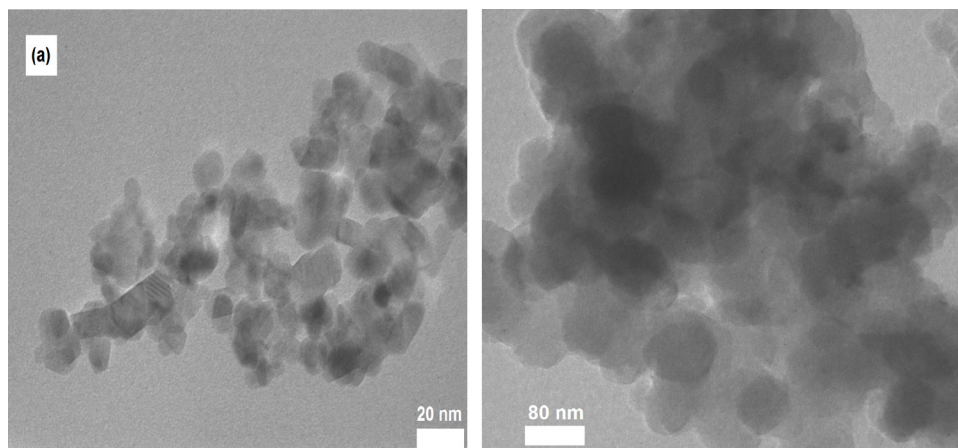


Fig. 5. (a) TEM image of SnO_2 prepared by ultrasound assisted. (b) TEM image of PPT/ SnO_2 nano hybrid prepared by ultrasound assisted chemical oxidative polymerisation precipitation.

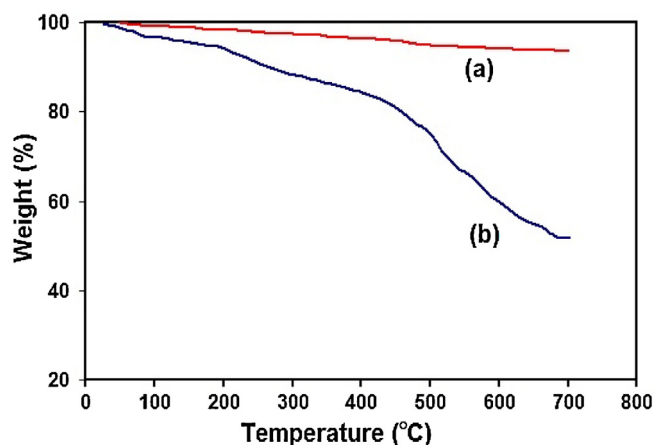


Fig. 7. Thermogravimetric analysis (TGA) of (a) SnO₂ and (b) PTP/SnO₂ nano hybrid.

chemical oxidative polymerization, enough time is not provided for the growth of particle size which leads to reduction in particle size. The turbulence generated due to the ultrasound induced acoustic streaming [52,53] also helps in preventing the large growth of particles or induces breakage giving lower final size. Also delayed polymerization time in the case of conventional chemical oxidative polymerization (i.e. 100 min) is responsible for larger particle size of PTP/SnO₂ hybrid.

3.4. Thermal gravimetric analysis (TGA) of SnO₂ and PTP/SnO₂ hybrid nanocomposites

The results of thermogravimetric analysis (TGA) of nanosize SnO₂ and PTP/SnO₂ have been given in Fig. 7. It has been observed that the nanoparticles of SnO₂ (Fig. 7(a)) show the first step loss in weight at temperature less than or equal to 150 °C. It may be due to the removal of physically adsorbed water and volatile impurities present on SnO₂ nanoparticles. In the second step chemically adsorbed water is removed from the system at temperature greater than 150 °C and decomposition of NH₄⁺ occurs at 390 °C. During this stage, the weight loss occurs gradually. In the case of PTP/SnO₂ hybrid (Fig. 7(b)) gentle decomposition with 3% weight loss at 100 °C and 43% weight loss at 700 °C has been observed. Also at 200 and 400 °C, a higher weight loss is attributed to the possible degradation of the polythiophene.

The SnO₂ particles were comparatively stable in the range of 0–700 °C. The cavitation micro jets facilitate the formation of crystalline SnO₂ particles during precipitation, which has been confirmed from the X-ray diffraction analysis. The major weight loss in TGA curve of hybrid composite was due to PTP degradation. The binding between the polymer and SnO₂ shows the strong chemical interaction as inferred from FTIR results [54,55].

3.5. Gas sensing study of polythiophene–SnO₂ hybrid nanocomposites

The room temperature gas response to various concentrations of LPG gas was measured by recording the resistance of the pellet in air and in the presence of test gas. The initial resistance of the PTP sensor was observed in the range of several kilo-ohms, while the PTP/SnO₂ sensor showed a lower resistance. Thus, the PTP/SnO₂ hybrid nanocomposite showed an increased conductivity compared to the pure PTP film, inferring that an increase in the conjugation length in PTP chains and the effective charge transfer between PTP and SnO₂ may result in an increase in the conductivity [54–57]. In addition, the improved contact between each grain via grain boundary may enhance the conductivity [58]. The

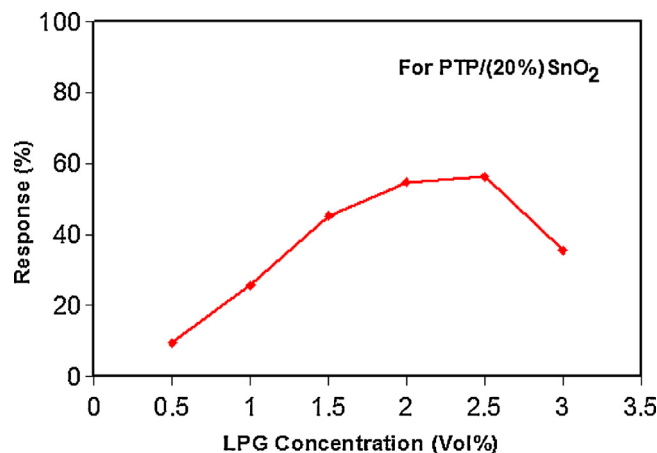


Fig. 8. Response of PTP/SnO₂ sensor as a function of the concentration of LPG gas (for PTP/(20%) SnO₂).

LPG response of the PTP/SnO₂ hybrid at an applied potential of +2 V is depicted in Fig. 8. From this figure, it is concluded that the gas response is a function of LPG concentration. The gas response increased from 9.5% to 56.2% with an increase in concentration of LPG from 0.5 to 2.5 vol%. The relative response tends to increase initially with increasing concentrations possibly due to the availability of a large number of reactive species in the sensing layer. However, the slow increase in response at higher concentration may be due to limited availability of surface area with possible reaction sites on the surface of the film due to adsorption of the gas molecules [59]. The maximum gas response of 56.2% was observed at 2.5 vol%. The enhanced response of PTP/SnO₂ hybrid nanocomposite can be explained on the basis of the fact that p-type (polythiophene) and n-type SnO₂ may form a p–n junction. Also, the creation of positively charged depletion layer on the surface of n-type material (SnO₂) due to the inter particle electron migration from n-type (SnO₂) to p-type (PTP) at the heterojunction helps in yielding better response characteristics. The net result would be in terms of the reduction of the activation energy and enthalpy of physisorption for LPG [60–62]. At 3 vol% of LPG, the response decreased to 35.6%. This may be due to the formation of multilayer of LPG molecules on the interface of junction at the higher gas concentrations resulting in the saturation in gas response.

The response/recovery time is an important parameter used for characterizing a sensor. The response time is defined as the time taken by the sensor to attain 90% of the maximum increase in the resistance on exposure to the target gas. Similarly, the recovery time can be defined as the time to get back 90% of the maximum resistance when exposed to air. The variation of response and recovery times with different concentration of LPG at room temperature has been given in Fig. 9. It has been observed that the response time decreases from 196 to 94 s, while recovery time increases from 182 to 466 s with increasing LPG concentration from 0.5 to 2.5 vol%. The observed trends may be attributed to the presence of sufficient gas molecules at the pellet surface of nanocomposite for the reaction to occur. From the same graph, it is found that at higher concentrations of LPG, the recovery time was long. This may be due to the higher density of LPG and the fact that the reaction products do not leave the interface immediately after the reaction.

Fig. 10 indicates that the presence of SnO₂ nanoparticles significantly reduces the response time of PTP sensor for 2 to 2.5 vol% of LPG concentration. It can be seen that the response time is dependent on the SnO₂ concentration in the PTP/SnO₂ sensors. The response (at 2–2.5 vol% LPG) of the neat PTP sensor was recorded within 410–360 s, which is reduced to 340–310, 280–224, 156–110 and 124–94 for the PTP/SnO₂ sensor containing 5, 10, 15 and 20

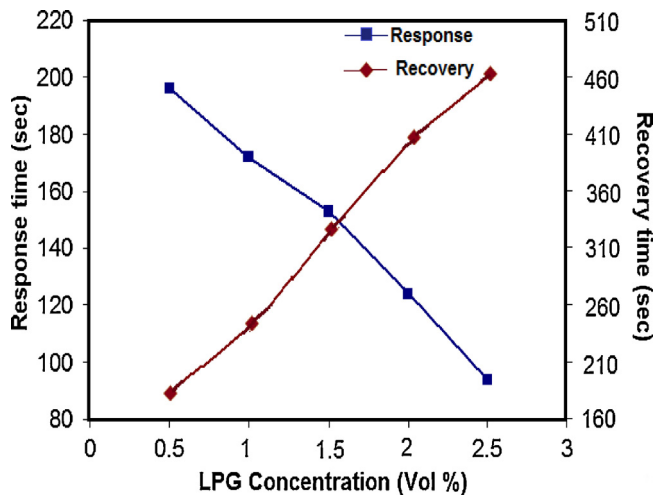


Fig. 9. Variation of response and recovery time of the PTP/SnO₂ hybrid sensor with LPG concentration (for PTP/(20%) SnO₂).

wt% of SnO₂, respectively. Thus, it can be established that the PTP/SnO₂ is more sensitive to LPG as compared to pure PTP. It indicates that different degree of interaction occurs between adsorbed LPG molecules and different sensors under the same experimental conditions. At higher concentrations of LPG, the diffusion rate is higher and sufficient quantities of gas molecules are available which reduces the response time.

3.5.1. LPG sensing mechanism of hybrid SnO₂/PTP

As already mentioned, pure PTP is a p-type semiconductor and the adsorption of LPG at PTP surface donates electrons and consequently its resistance increases. On the other hand, SnO₂ is an n-type semiconductor and its resistance decreases, when exposed to the reducing gas such as LPG. When hybrid PTP/SnO₂ nanocomposite is exposed to LPG, it exhibits the properties of p-type semiconductor, that is, its resistance increases on exposure to LPG. This observation suggests that the LPG sensing mechanism of the hybrid is governed by PTP, which is supported by TEM study. As revealed by TEM study, the SnO₂ nanoparticles are covered and interconnected with PTP in the nanocomposite and therefore, the active sites on the PTP backbone are largely accessible to LPG [62,63]. The response time of hybrid toward LPG is considerably lower than that of the pure PTP. The hybrid exhibits excellent LPG sensing characteristics than pure PTP, suggesting that the SnO₂

nanoparticles in the nanocomposite are also playing an active role in the sensing process.

The schematic representation of the model based on the principle of formation of heterojunction barrier is depicted in Fig. 11 to explain the LPG sensing behavior of hybrid PTP/SnO₂ nanocomposite. When, the hybrid nanocomposite is exposed to the LPG, the LPG gas molecules are adsorbed preferentially on PTP and it leads to the interaction of LPG at the interface of p-PTP/n-SnO₂ heterojunction. The adsorption of LPG at PTP surface donates electrons and results in more negative acceptor ions on the p-side and hence there is a possibility of a wider space charge layer. As the surface of SnO₂ nanoparticles is generally covered with negatively charged adsorbed O⁻ ions, an electron depletion layer is created on n-side of the junction. As a result, the carrier concentration of the heterojunction decreases and consequently, the potential barrier height of the heterojunction increases giving an increase in the resistance of the hybrid. Therefore, PTP identify LPG (receptor function) and both PTP and SnO₂ nanoparticles provide conducting path (transducer function) and hence the nanocomposite permits selective detection of LPG at room temperature.

3.5.2. Comparison of hybrid with other sensors

The combination of these two materials (conducting polymer and SnO₂) to give a hybrid nanocomposite can lead to a synergistic effect, which are helpful for improving the sensor properties such as thermal stability, lower operating temperature, and fast response and recovery. To establish the utility of hybrid sensors, it is essential to compare the response characteristics of the hybrid sensor with some of the reported literature illustrations.

Song et al. [62] synthesized polypyrrole/ZnSnO₃ nanocomposite which were prepared by a simple in situ chemical polymerization method. It has been reported that these nanocomposites exhibited a higher response upto 90% for 1000 ppm NH₃ gas. Xu et al. [49] reported that the NO₂ sensor based on PTP-coated SnO₂ hollow spheres exhibited higher gas response and shorter recovery time for detecting NO₂ at ppm levels at 90 °C in comparison with the sensor based on pure PTP. Dhawale et al. [63] fabricated p-polyaniline/n-ZnO thin film heterojunction sensor for room temperature LPG detection via electrodepositing polyaniline on chemical bath deposited ZnO film. The heterojunction sensor has quick and high response toward LPG as compared to N₂ and CO₂ and exhibited maximum response of 81% upon exposure to 1040 ppm of LPG. Deshpande et al. [64] synthesized the SnO₂-intercalated PANI nanocomposite, which showed better sensitivity to ammonia gas at room temperature than SnO₂ with the response and recovery times of 12–15 s and 80 s for 300 ppm ammonia gas, respectively. However, pure PANI exhibited better sensitivity to ammonia than SnO₂-intercalated PANI nanocomposite with relatively fast response (8–10 s) and slow recovery (160 s). Kong et al. [65] recently reported the synthesis of SnO₂-PTP composites and studied their NO₂ sensing properties at low operating temperatures, which were reported to be better than SnO₂. Patil et al. [66] investigated Poly(*o*-anisidine)-tin oxide (POA-SnO₂) nanocomposites via in situ chemical polymerization of *o*-anisidine in presence of SnO₂ nanoparticles. The POA-SnO₂ nanocomposite shows better LPG sensing properties than that observed for the pure POA. The nanocomposite with 50 wt% SnO₂ exhibit excellent LPG sensing characteristics at the operating temperature of 100 °C such as higher relative gas response (~23.47% to 3.4% of LPG), extremely rapid response (~6 s), fast recovery (~33 s), good reproducibility, and remarkable selectivity. In contrast, the present LPG sensor based on PTP/SnO₂ hybrid has a relative LPG response of 56.2% for 2.5 vol% LPG at room temperature as shown in Fig. 8. Also it has a response time of 94 s and recovery time of 466 s for 2.5 vol% at PTP/(20%) SnO₂, respectively. Also PTP/SnO₂ hybrid nanocomposites have better LPG sensing characteristics than the

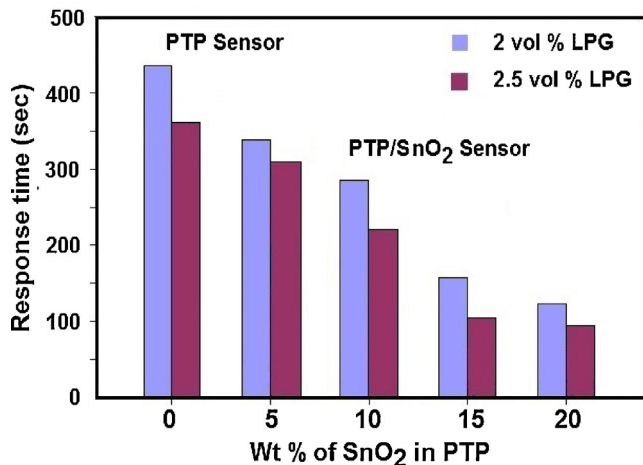


Fig. 10. Variation of response time as a function of different wt% of SnO₂ in the sensor at different LPG concentrations.

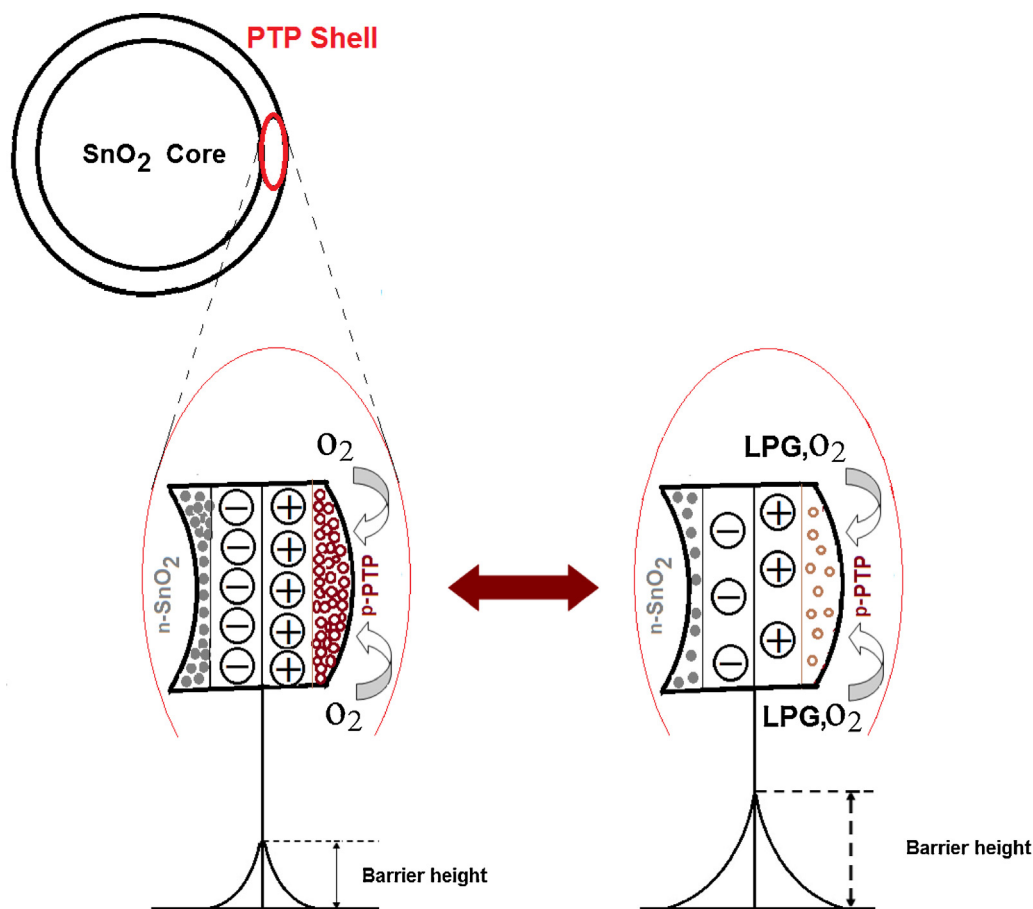


Fig. 11. Schematic representation of LPG sensing model for PTP/SnO₂ nano hybrid.

semiconducting metal oxide based LPG sensors reported in the literature [66–70] and can overcome the shortcomings of longer response time, slow recovery and lack of selectivity of PTP and requirement of higher operating temperature of SnO₂, thus presenting very promising sensing material for the fabrication of LPG sensors operating at low temperatures.

4. Conclusions

Inorganic-organic hybrid materials of SnO₂ nanoparticles coated by PTP have been successfully prepared through an ultrasound assisted in situ chemical oxidative polymerization of thiophene in the presence of SnO₂ nanoparticles. The presence of ultrasonic irradiation creates conditions suitable for intense micromixing resulting into lower sizes of the final composite as established using comparisons with experiments based on the conventional approach in the absence of ultrasound. XRD, FTIR and TEM results showed that there is a certain synergetic interaction between the SnO₂ nanoparticles and polythiophene. Uniform encapsulation of SnO₂ particles into the PTP improves the LPG sensing performance, implying the potential sensor applications of the hybrids. The maximum sensing capability along with better long term stability was observed for the sensor containing 20 wt% SnO₂.

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